## **Direct** growth of WS<sub>2</sub> on graphene and other substrates towards functional hetero-stacks

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Both profound research into two-dimensional (2D) materials and their use in applications is by no means limited to graphene. Instead, graphene, with its striking properties such as electron mobility and mechanical flexibility, marked the beginning of a new era in material science.<sup>1</sup>Novel emerging 2D materials might overcome some of the shortcomings of graphene for selected applications. Bv direct displaying a measurable bandgap, semiconducting transition metal dichalcogenides (TMDs) such as molybdenum disulfide (MoS<sub>2</sub>) and tungsten disulfide (WS<sub>2</sub>) offer exciting prospects for the development of novel electronic and optoelectronic devices. Indeed, opportunely designed 2D-heterostacks hold the potential to outperform conventional materials for a wide variety of applications<sup>2</sup> including flexible and transparent electronics with low power consumption - and might potentially push applications like the hydrogen evolution reactor over the barrier of sufficient effectivity for industrial fabrication<sup>3</sup>.

Up to recently,  $WS_2$  has attracted the second most attention within the TMD family, after  $MoS_2$ , but is now considered superior over  $MoS_2$  regarding several key properties such as a higher carrier mobility, a larger valence band splitting allowing for easier observation of the valley hall effect and a much higher photo



 $\label{eq:sigma} \begin{array}{ll} Fig. 1 \mid & Simultaneous \ CVD \ growth \ of \ WS_2 \ on \ various \\ & substrates. \end{array}$ 

**a**, Schematic experimental setup for CVD growth. The red marked zone corresponds to the hot  $(900^{\circ})$ zone within the furnace. **b-e**, Typical SEM images of the as-grown WS<sub>2</sub> crystals on graphene (**b**), SiC (**c**), SiO2 (**d**) and an hBN-flake (**e**).

luminesce emission.<sup>4</sup>Unlike Graphene, single sheets of  $WS_2$  do not exhibit the thickness of one atom but rather of one molecule, since a single atomic layer of W is sandwiched between two atomic layers of S. Strong covalent W-S bonds provide for a high intralayer stability, while inter-layer cohesion is produced by much weaker van der Waals forces only, therefore classifying  $WS_2$  as a 2D material.

- 1. Ferrari, A. C. *et al*.Science and technology roadmap for graphene, related two-dimensional crystals, and hybrid systems. *Nanoscale***7**, 4587–5062 (2015).
- Lim, H., Yoon, S. I., Kim, G., Jang, A. & Shin, H. S. Stacking of Two-Dimensional Materials in Lateral and Vertical Directions. *Chem. Mater.* 26, 4891–4903 (2014).
- 3. Yang, J. *et al.* Two-dimensional hybrid nanosheets of tungsten disulfide and reduced graphene oxide as catalysts for enhanced hydrogen evolution. *Angew. Chemie Int. Ed.***52**, 13751–13754 (2013).
- 4. Lan, C., Li, C., Yin, Y. & Liu, Y. Large-area synthesis of monolayer WS2 and its ambient-sensitive photo-detecting performance. *Nanoscale***7**, 5974–5980 (2015).
- 5. Miseikis, V. *et al.* Rapid CVD growth of millimetre-sized single crystal graphene using a cold-wall reactor. *2D Mater.***2**, 014006 (2015).
- 6. Starke, U., Forti, S., Emtsev, K. V & Coletti, C. Engineering the electronic structure of epitaxial graphene by transfer doping and atomic intercalation. *MRS Bull.***37**, 1177–1186 (2012).
- 7. Gutie, H. R. *et al.* Extraordinary Room-Temperature Photoluminescence in Triangular WS2 Monolayers. *Nano Lett.***13**, 3447–3454 (2013).
- 8. Berkdemir, A. et al. Identification of individual and few layers of WS2 using Raman spectroscopy. Sci. Rep.3, 1755 (2013).
- 9. We acknowledge our colleagues V. Miseikis, D. Convertino and N. Mishra for providing the graphene on exfoliated hBN, graphene on SiC and transfered graphene substrates.

To date, the synthesis of WS<sub>2</sub> still lags behind ongoing technological progress the of graphene regarding the fast and relatively cheap fabrication of large and high quality mono-crystalline layers<sup>5</sup>.  $WS_2$  flakes are typically obtained by mechanical exfoliation of 3D crystals and WS<sub>2</sub>-containing heterostacksare realized by a cumbersome process of mechanical assembly. The limited lateral dimension of the resulting samples restricts the access to conventional surfacescience techniques for fundamental studies and is a serious hurdle towards the implementation of a scalable technology. Clearly, significant advances in the scalable synthesis of highly crystalline WS<sub>2</sub> on a variety of substrates are called for.

In this work we present a cost-effective vaporphase reaction approach which allows for the simultaneous deposition of WS<sub>2</sub>on a variety of substrates ranging from classical 3D dielectrics to novel 2D materials. The quality, crystallinity and thickness of the deposited layers is thoroughly characterized via Raman spectroscopy, scanning electron microscopy (SEM) and atomic force microscopy (AFM).

In Fig. 1(a), we illustrate schematically our typical growth setup. A large diameter quartz tube is placed within a LENTON furnace. Inside the tube, we position a second semisealed tube of smaller diameter whose purpose is to create a semi-confined volume of reactants for an increased reaction cross section. In the hot zone, the tungsten trioxide  $(WO_3)$  precursor is placed next to the receiving substrate(s) within a quartz crucible. The sulfur (S) precursor is placed just in front of the open end of the small diameter tube outside of the furnace. Argon gas (Ar) is flown with a rate of 80-200 sccm as a carrier gas from one end, while a roughing pump on the other side caters for a pressure of a few mbar. After 10 min of pumping and purging with Ar flow, the temperature is first raised at a rate of 10°C per minute to 400°C and then at a rate of 45°C per minute until the growth temperature of 900°C is reached. The growth time is fixed at 90 minutes followed by a natural cool down.

Panels (b-e) in Figure 1 report SEM micrographs of single crystals of WS<sub>2</sub> obtained



Fig. 2 | Verification of monolayer WS<sub>2</sub> depositions. a, Strong photoluminescence response of WS<sub>2</sub> on SiO<sub>2</sub> to a 1200nm excitation. b, Map showing the intensity of the 340-360 nm Raman signal (characteristic for WS<sub>2</sub>) overlaid over an optical image. c, AFM images of signal amplitude (profile along blue line shown in inset) and d, signal phase.

on silicon carbide (SiC), silicon dioxide  $(SiO_2)$ ,graphene, and hexagonal boron nitride (h-BN), respectively. On all substrates we obtain triangular grains of m size, with the largest crystals (>5 m) on SiO<sub>2</sub>.

In all cases we confirmed the thickness of the vast majority of the deposited TMD to be a single layer via AFM analysis. The AFM topography image and line profile reported in Fig. 2(c) highlight the monolayer thickness (~1nm) of the triangular  $WS_2$  mono-crystals obtained on epitaxial graphene on  $SiC^6$ . The phase AFM image in panel (d)clearly evidences chemical the and physical differentness from the underlying graphene photoluminescence Thestrong terraces. measured for WS<sub>2</sub> on SiO<sub>2</sub>(due to its direct band  $gap^7$ ) is reported in Fig. 2(a). In Fig. 2b, we present an optical micrograph of  $WS_2$ grown on CVD graphene<sup>5</sup> transferred to SiO<sub>2</sub>with a superimposed map showing the intensity of the Raman response between 340 and 360nm, which has been identified as a fingerprint for  $WS_2^8$ .

In summary, in this work we report a robust growth process for the simultaneous synthesis of single-crystal m-sized monolayer WS<sub>2</sub> on various substrates<sup>9</sup> andthoroughly characterize the synthetized TMD by using different spectroscopic and microscopic techniques.